

Applications of XAFS to materials and nano – science

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Plan

- Examples of applications, using both
 - results which have “stood the test of time”
 - recent results



Today's topics

- Dopants, defects
- Alloys
- Phase transitions
- Thin films, interfaces
- Nanostructures



XAFS and dopants/dilute elements

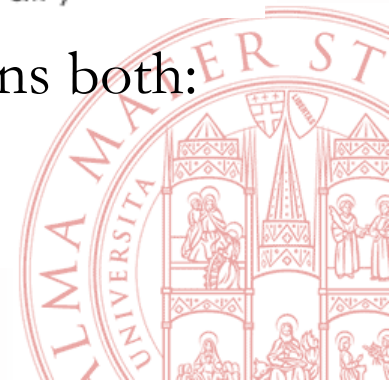
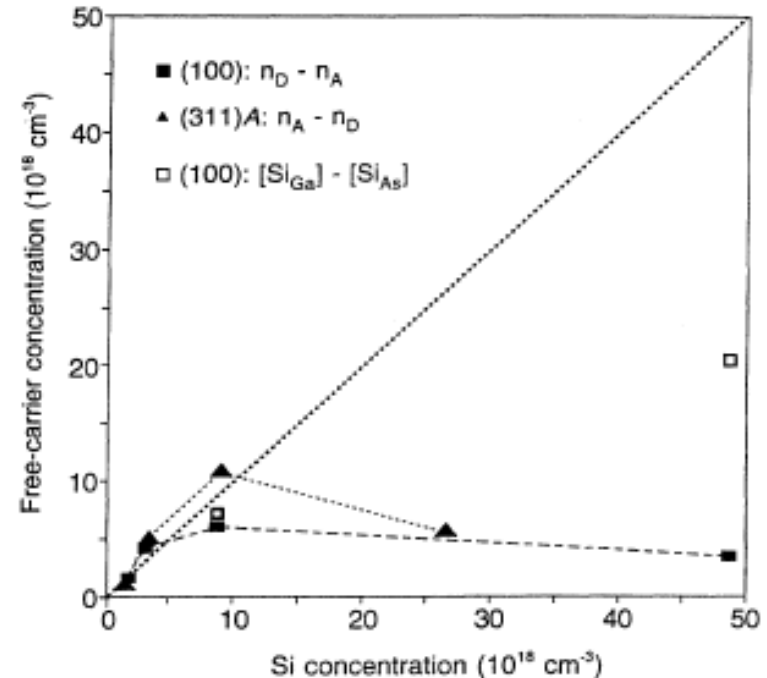
- Only the structure around the photo-excited atom is probed
- Fluorescence detection greatly enhances sensitivity
- Present sensitivity limit (depends on sample)
 - dopants in the bulk
 - EXAFS $\sim 10^{18}$ at/cm³
 - XANES $\sim 10^{17}$ at/cm³
 - thin films (single layer) $\sim 10^{14}$ at/cm²



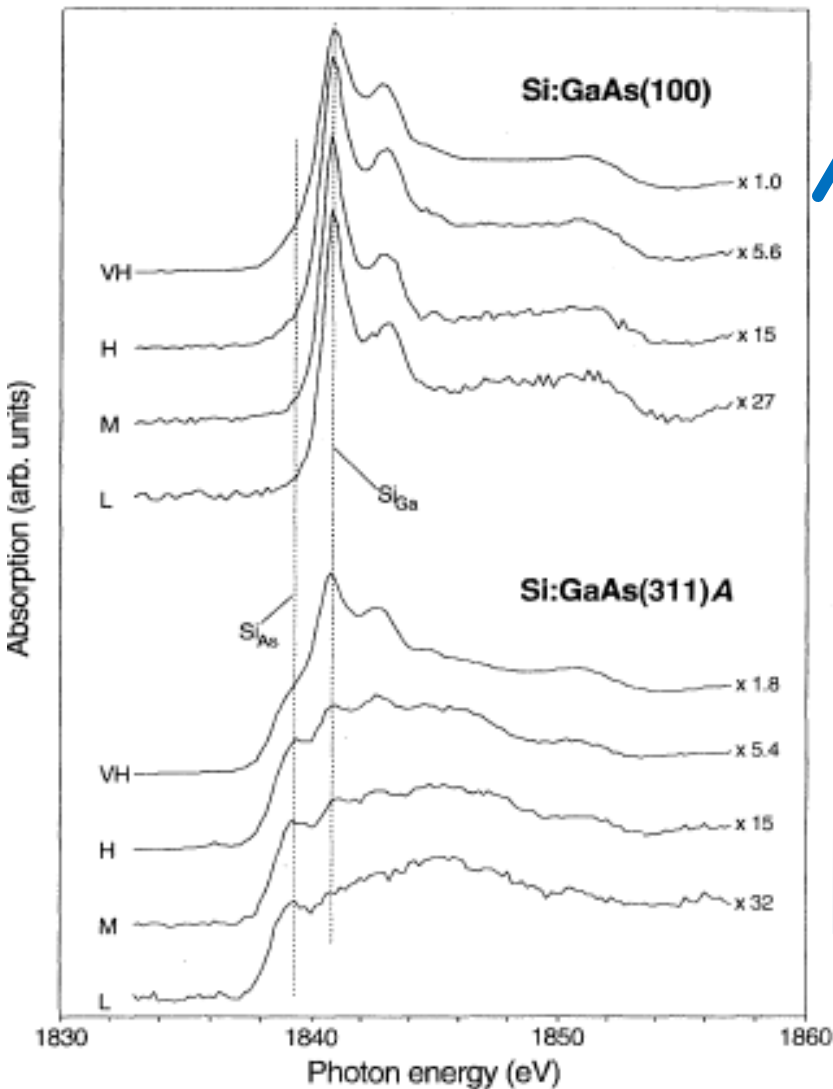
Si in GaAs

S. Schuppler, D.L. Adler, L.N. Pfeiffer, K.W. West, E.E. Chaban,
and P.H. Citrin, Appl. Phys. Lett. **63**, 2357 (1993) and Phys. Rev. B **51**, 10527 (1995)

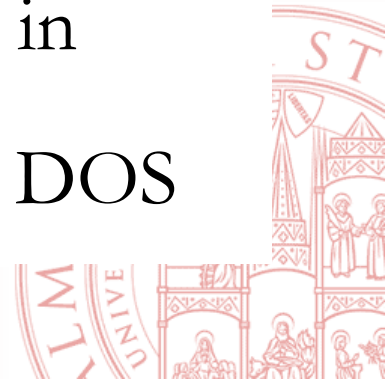
- Si common dopant in GaAs
- Si:GaAs exhibits deactivation
- Accepted explanation : amphoteric nature of Si
 - Si_{Ga} (Si in Ga site): donor
 - Si_{As} : acceptor
 - At low concentration all Si_{Ga} , at higher concentrations both: autocompensation



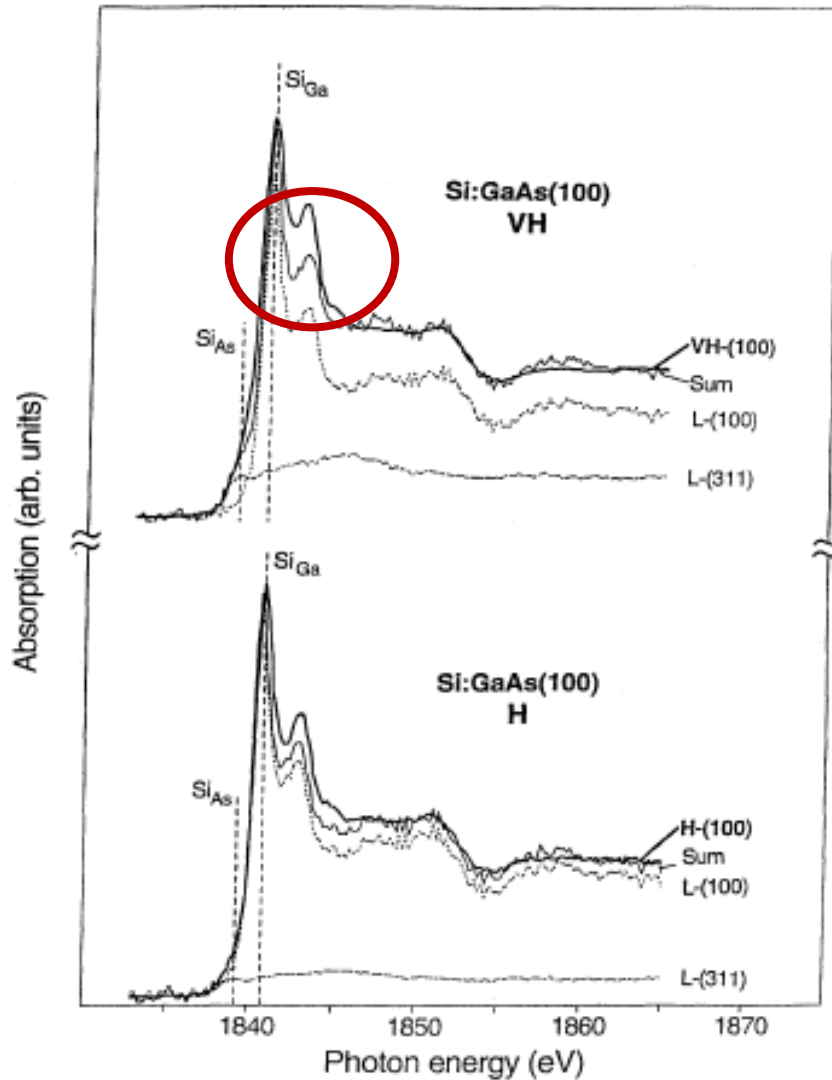
Si in GaAs: XANES



- Samples studied
 - Si:GaAs(001)
 - at low concentration Si_{Ga}
 - Si:GaAs(311)A
 - at low concentration Si_{As}
- XANES exhibit reasonable evolution with concentration
- Difference in lineshape between Si_{Ga} and Si_{As} due to difference in charge on Si and conduction band DOS



Si in GaAs: XANES

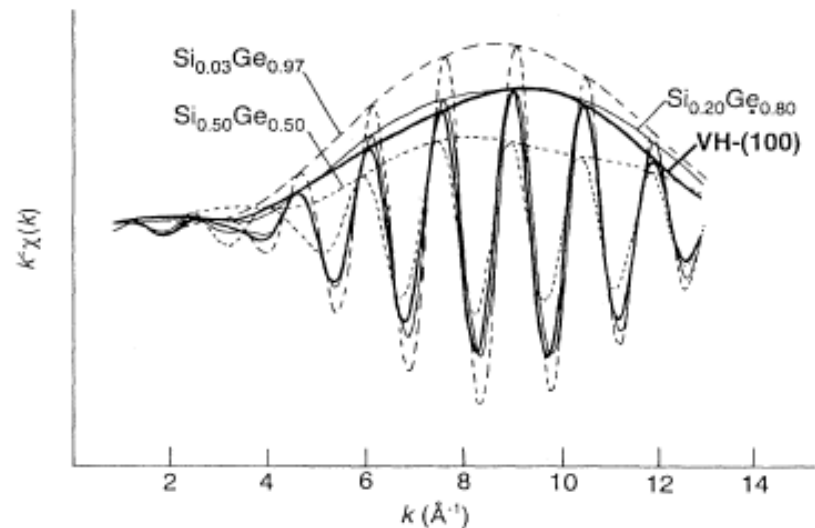
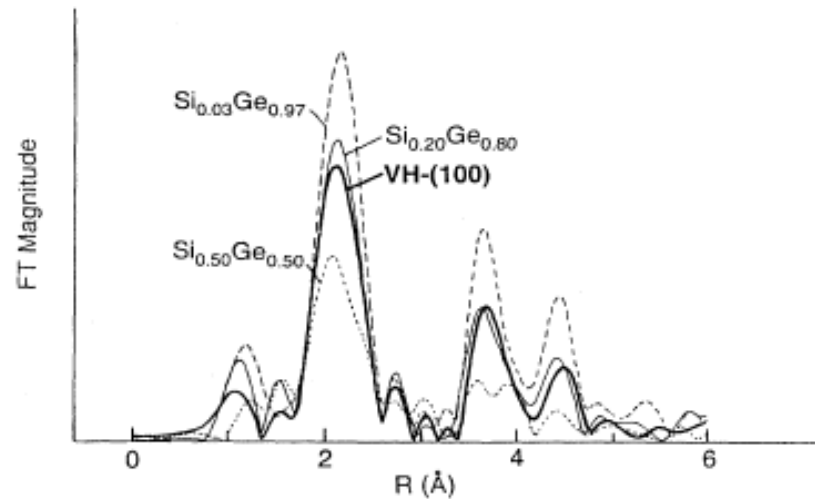


- Fitting of Very High concentration sample indicated that lineshape cannot be explained only on the basis of combination of Si_{Ga} and Si_{As}



Si in GaAs : EXAFS

- Compare EXAFS spectra with those of $\text{Si}_x\text{Ge}_{1-x}$ random alloys
- Ge has similar scattering amplitude to Ga and As
- VH sample spectrum very similar to $\text{Si}_{0.2}\text{Ge}_{0.8}$
 - 20% of Si is bonded to Si
- Conclusion: deactivation due also to presence of **Si dimers and clusters**



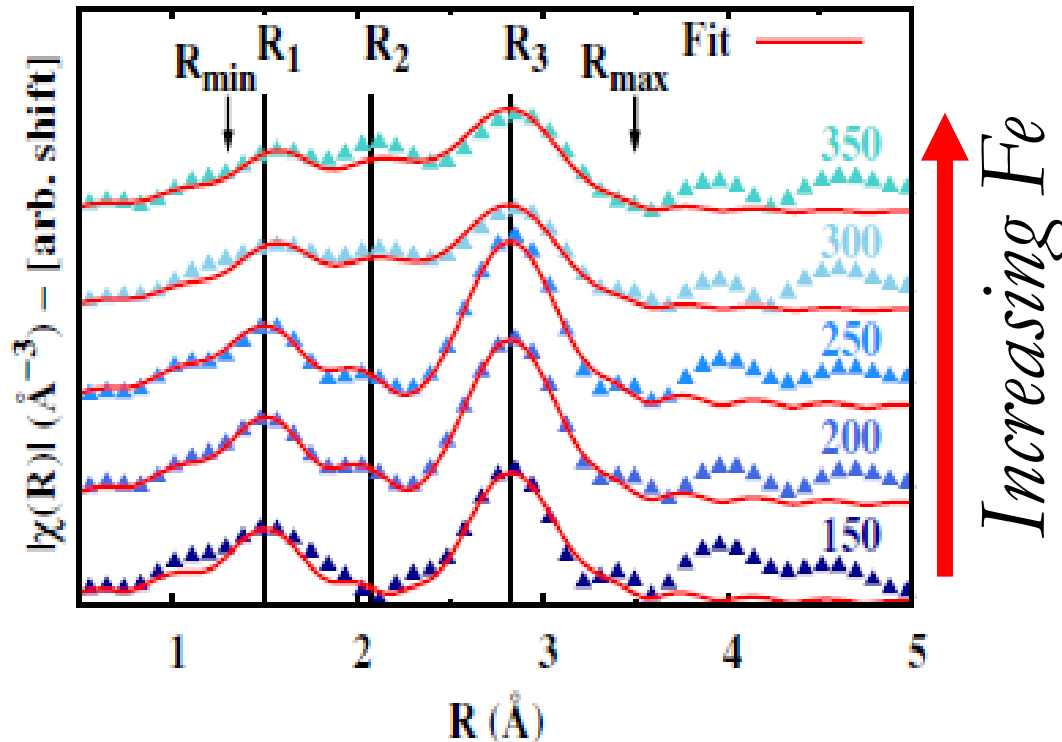
Fe in GaN

- A. Bonanni, A. Navarro-Quezada, Tian Li, M. Wegscheider, Z. Matěj, V. Holý, R. T. Lechner, G. Bauer, M. Rovezzi, F. D'Acapito, M. Kiecana, M. Sawicki, and T. Dietl, Phys. Rev. Lett. **101**, 135502 (2008)
- M. Rovezzi, F. D'Acapito, A. Navarro-Quezada, B. Faina, T. Li, A. Bonanni, F. Filippone, A. Amore Bonapasta, and T. Dietl, Phys. Rev. B **79**, 195209 (2009)
- A. Navarro-Quezada, W. Stefanowicz, Tian Li, B. Faina, M. Rovezzi, R. T. Lechner, T. Devillers, F. D'Acapito, G. Bauer, M. Kiecana, M. Sawicki, T. Dietl, and A. Bonanni Phys. Rev. **81**, 205206 (2010)

- Candidate material for spintronic applications
- Grown by Metal Organic Vapor Phase Epitaxy
- Fe concentrations
 $4 \times 10^{19} \text{ cm}^{-3} - 4 \times 10^{20} \text{ cm}^{-3}$
- Aims:
 - Determine the site of Fe in GaN
 - Determine the effect of Si co-dopant
 - Correlate with magnetic properties



Fe:GaN data



Low Fe content:

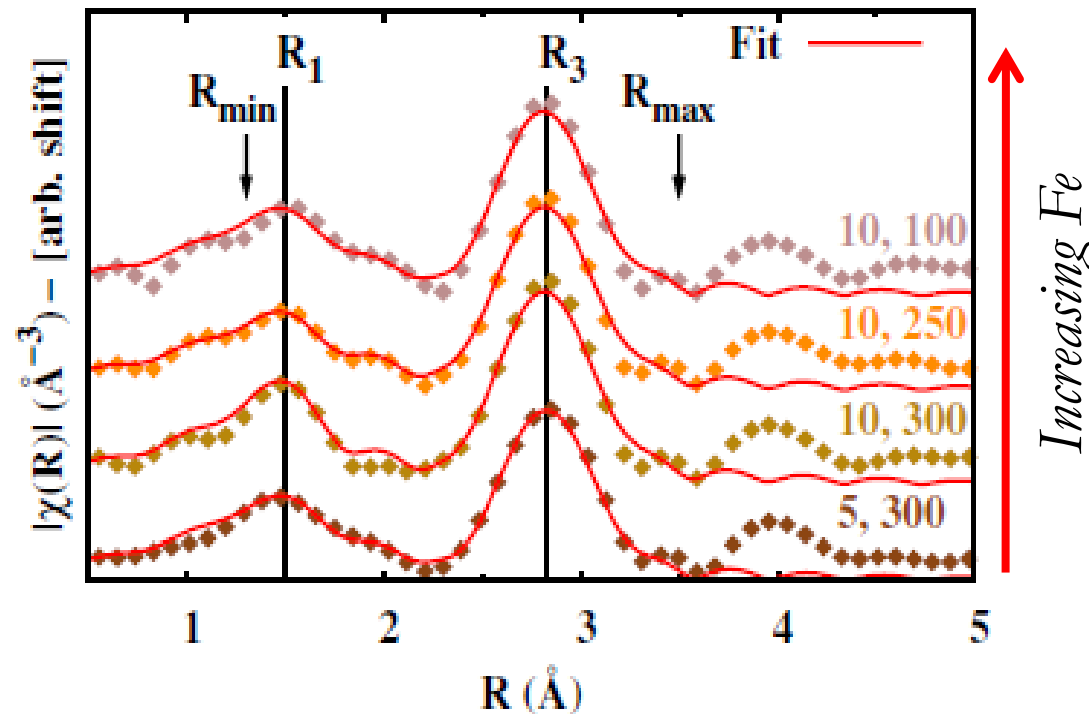
- only two Fe-N (R_1) and Fe-Ga (R_3) bonds.
- Fe substitutional; bond length in agreement with DFT for Fe^{3+}

High Fe content

- Appearance of Fe-Fe (R_2) coming from a precipitated Fe_xN phase



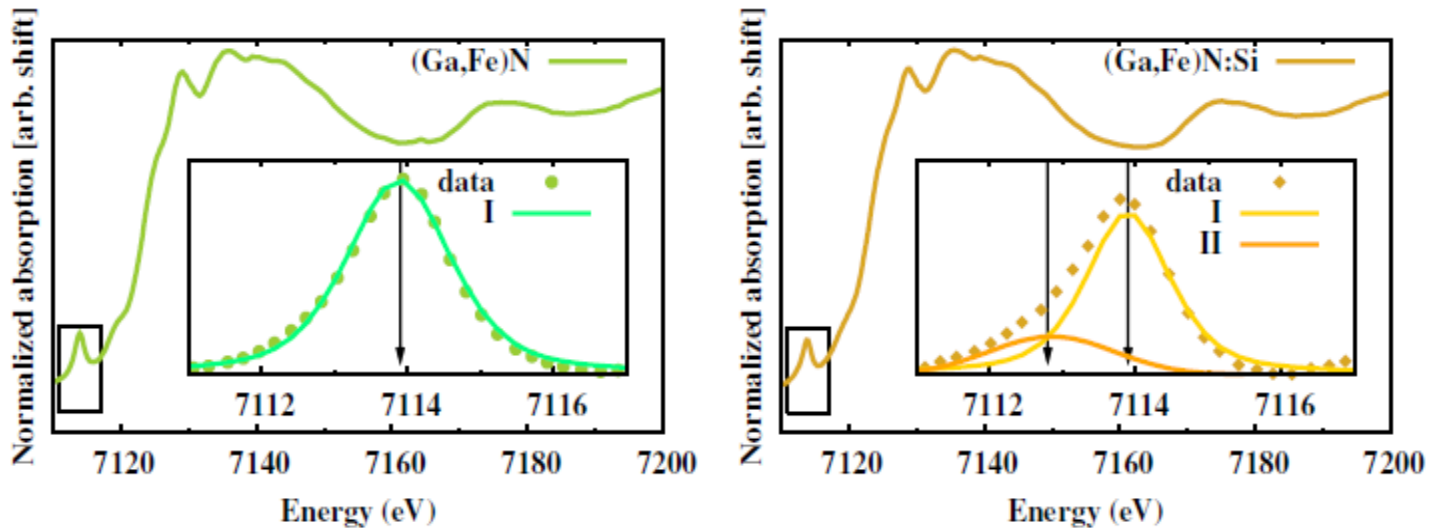
Si,Fe:GaN data



- For the same Fe content Si co-doping prevents the formation of Fe_3N
- No evidence of the Fe-Fe bond at R_2



Si affects the charge state of Fe



- Si addition causes partial reduction of Fe^{3+} ions to Fe^{2+}
- Notable ability of XAFS to determine structure and valence



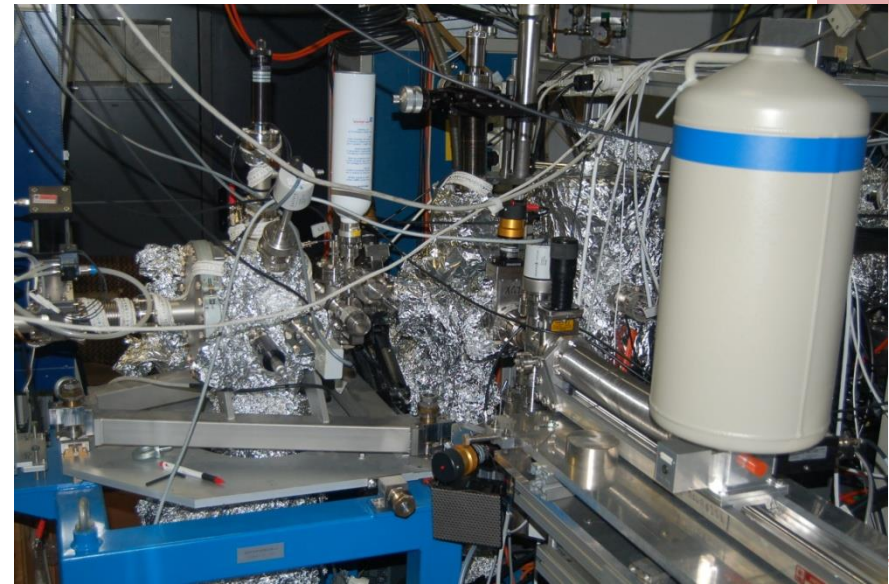
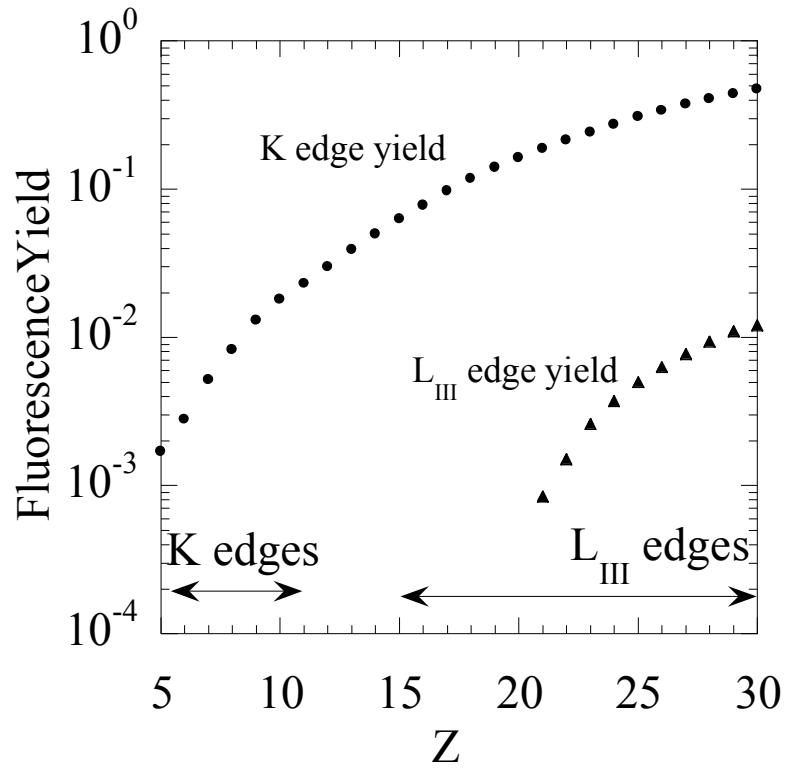
Fe:GaN conclusions

- Magnetization due to various components, including one due to ferromagnetic nanocrystals of ϵ -Fe₃N, α -Fe, γ' -Fe₄N, γ -Fe₂N and γ -Fe
- Si codoping reduces the formation of Fe rich nanocrystals and permits a higher incorporation of Fe.
- Use new term: (Ga,Fe)N nanocomposites, not real doping



Low Z dopants and XAS

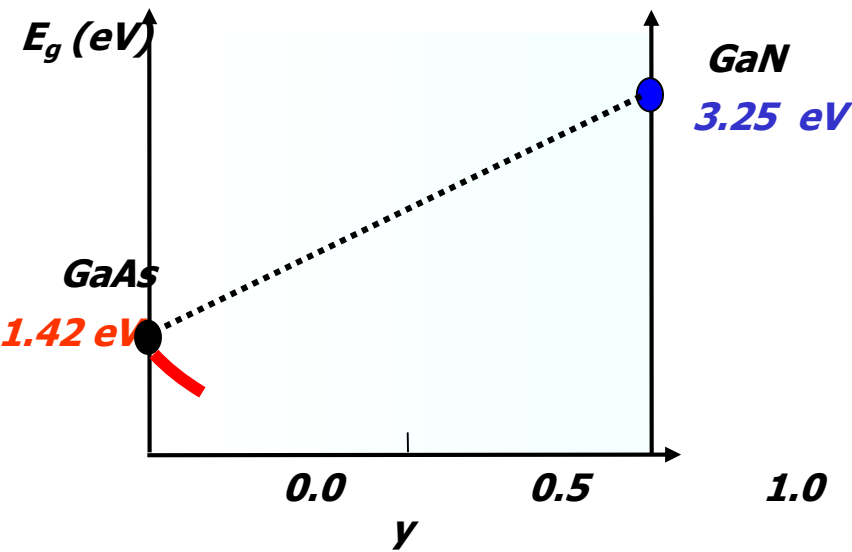
- C, N & O often used as dopants
- Experimentally difficult: low fluorescence yield, soft X-rays, UHV



ALOISA beamline @ ELETTRA



Dilute nitrides: $\text{GaAs}_{1-y}\text{N}_y$, $\text{In}_x\text{Ga}_{1-x}\text{As}_{1-y}\text{N}_y$

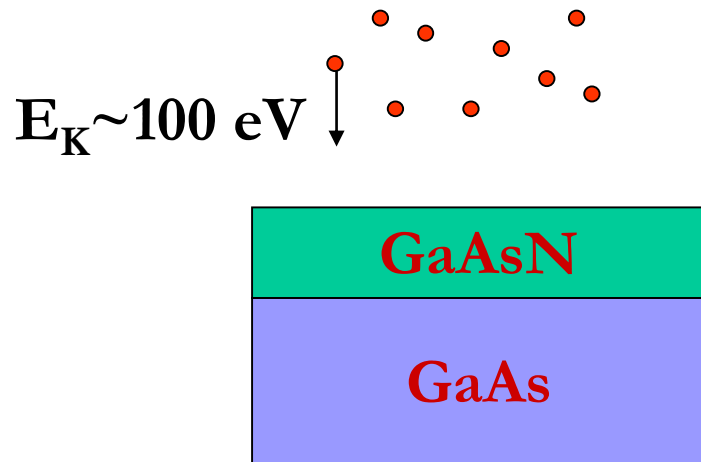


- Anomalous non-linear optical and electronic properties of III-V nitrides
- Red shift of the band gap by adding few % of nitrogen (≈ 0.05 - 0.1 eV per N atomic percent in InGaAsN)
- Huge and composition dependent optical bowing

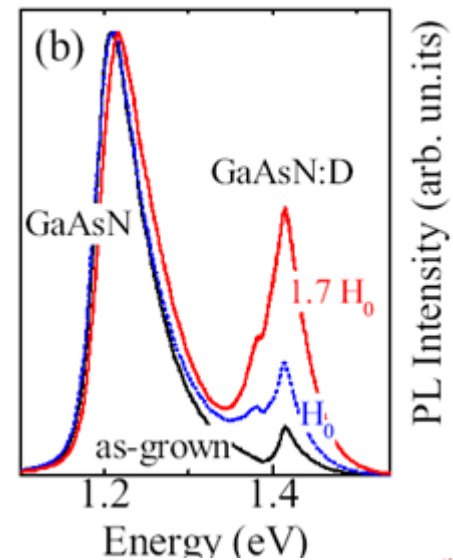


Hydrogen – nitrogen complexes in dilute nitrides

- Hydrogenation leads to reversible opening of E_g



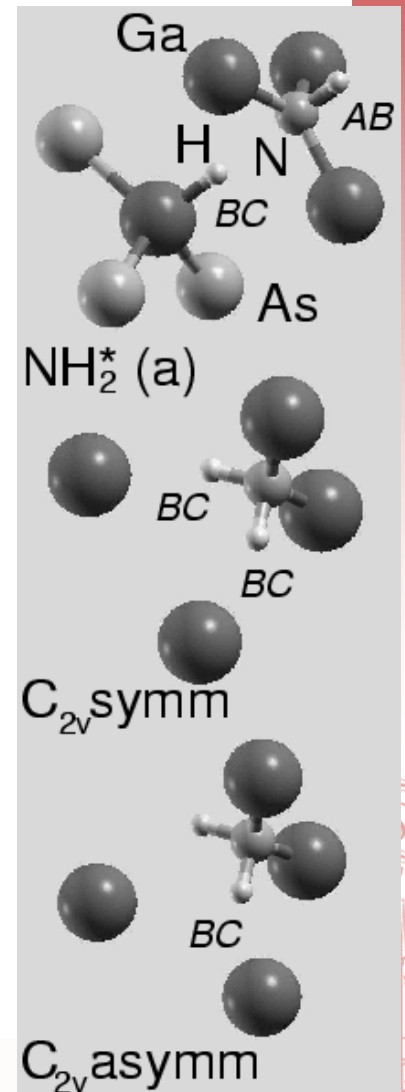
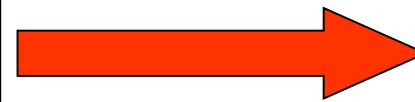
photoluminescence



Hydrogen – nitrogen complexes in dilute nitrides

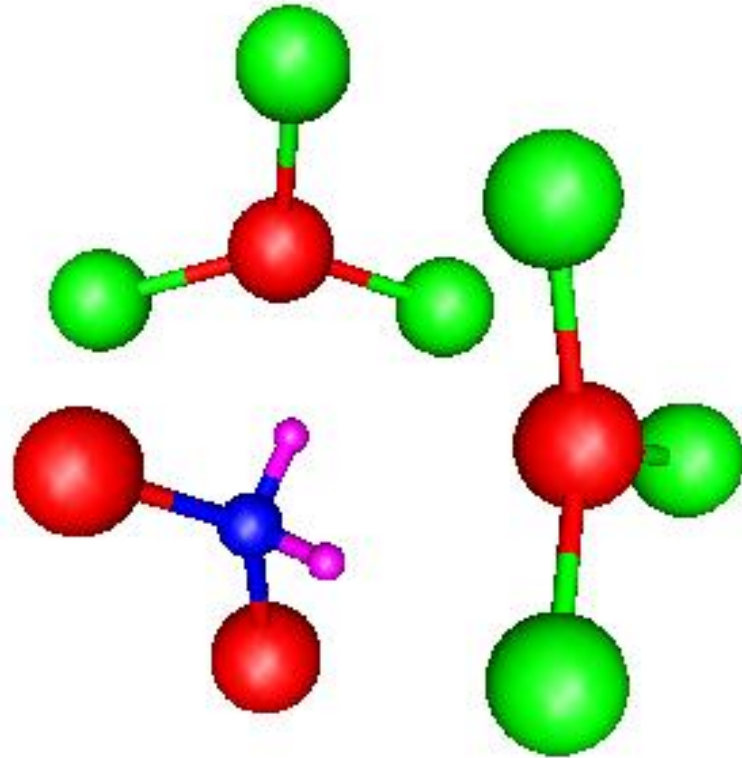
- Which is the hydrogen –nitrogen complex responsible for these changes?

Some candidate
low energy structures



H–N complexes in dilute nitrides

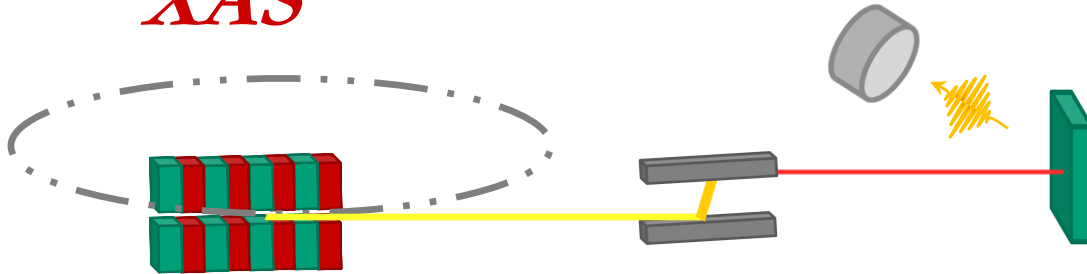
- G. Ciatto, F. Boscherini, A. Amore Bonapasta, F. Filippone, A. Polimeni and M. Capizzi, Phys. Rev. B **71**, 201301 (2005)
- M. Berti, G. Bisognin, D. De Salvador, E. Napolitani, S. Vangelista, A. Polimeni, M. Capizzi, F. Boscherini, G. Ciatto, S. Rubini, F. Martelli, and A. Franciosi, Phys. Rev. B **76**, 205323 (2007)
- G. Ciatto, F. Boscherini, A. Amore Bonapasta, F. Filippone, A. Polimeni, M. Capizzi, M. Berti, G. Bisognin, D. De Salvador, L. Floreano, F. Martelli, S. Rubini, and L. Grenouillet, Phys. Rev. B **79**, 165205 (2009)
- DFT calculations to determine lowest energy geometries
- Full multiple scattering XANES simulations
- **Answer:** C_{2v} – like complexes are mostly present
- 3-D sensitivity of XANES!!



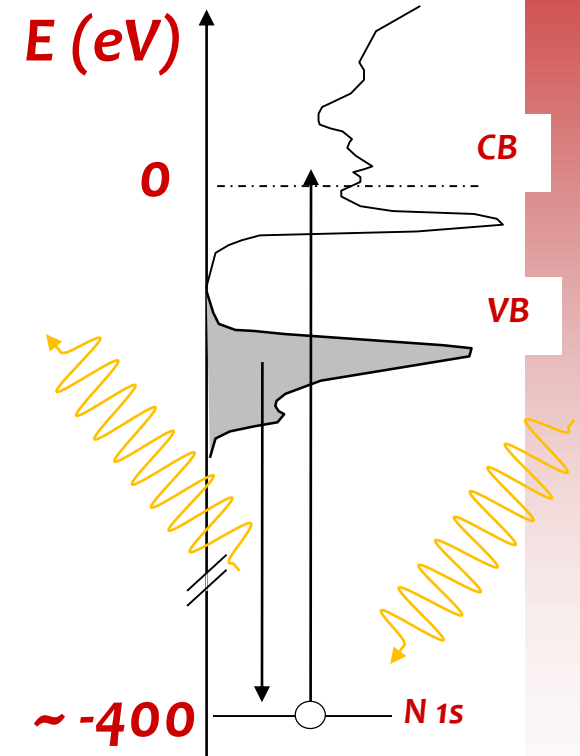
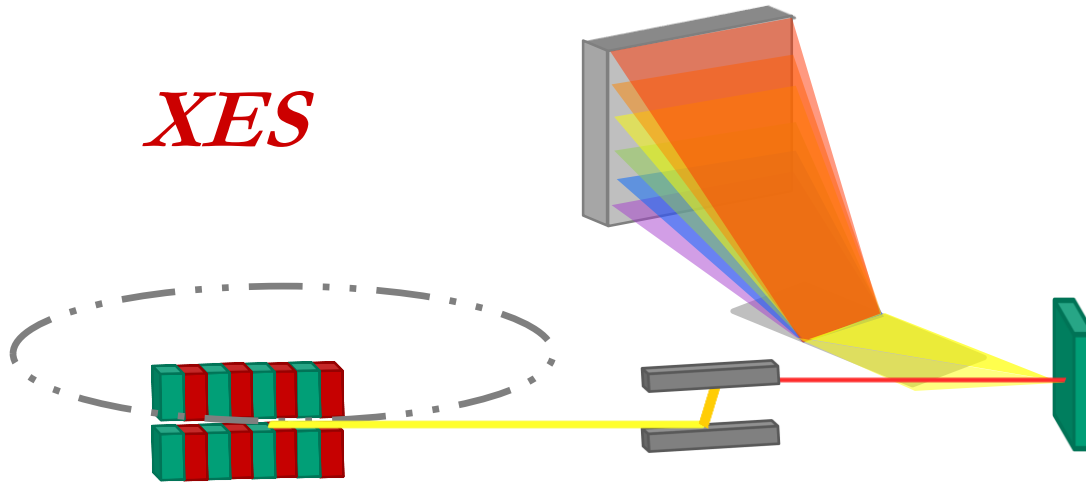
Combined XAFS and XES

Amidani et al., Phys. Rev. B 89, 085301 (2014), talk MS 103.O05

XAS



XES



XES now possible, a complementary tool with sensitivity to

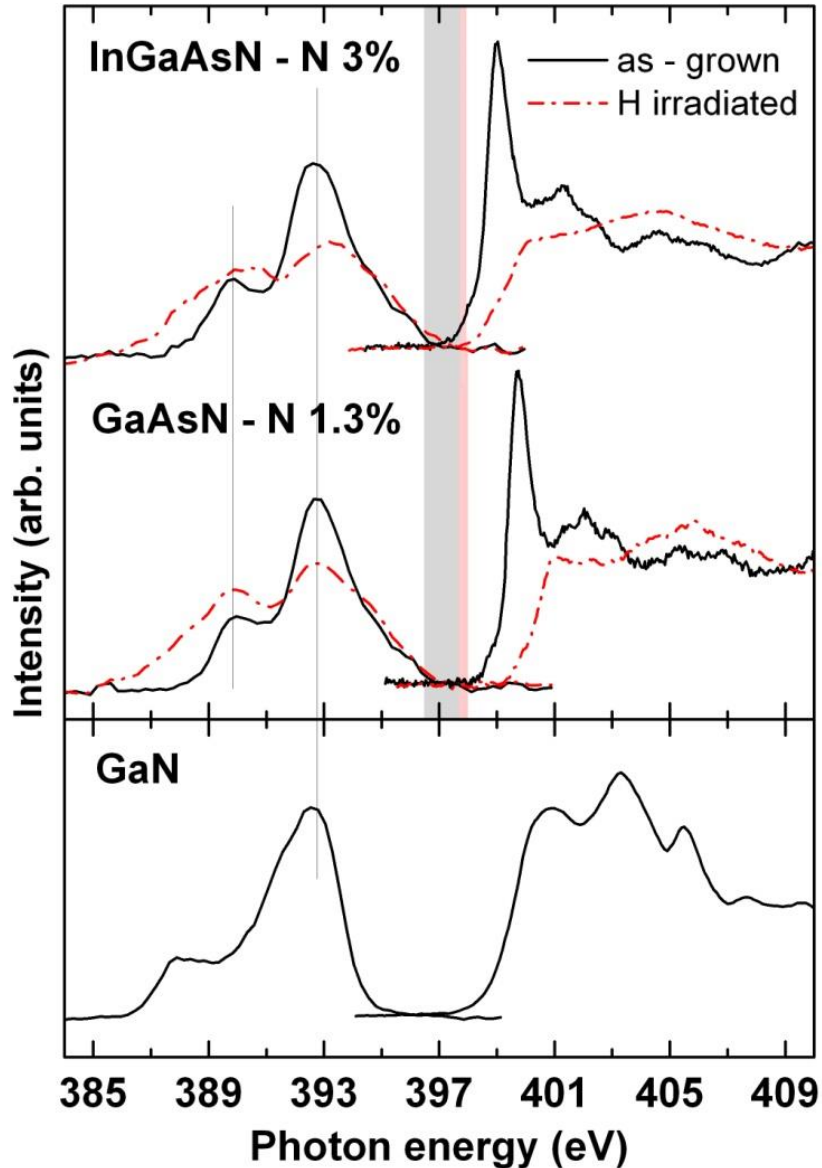
- Valence band electronic structure
- Atomic structure
- New level of refinement in x-ray spectroscopy



Combined XAFS and XES

XES

- local VBM unchanged
- decrease of main peak in favor of lower energy states

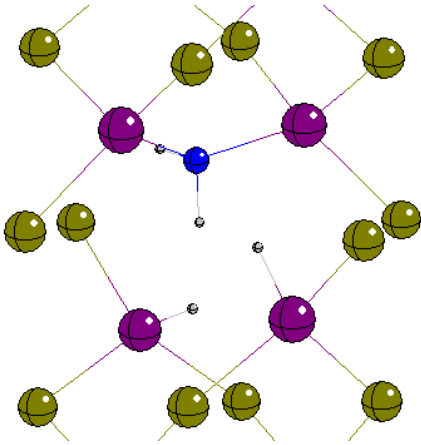


XAFS

- main peak disappears and local CBM is strongly blue-shifted
- N states move far from the CBM

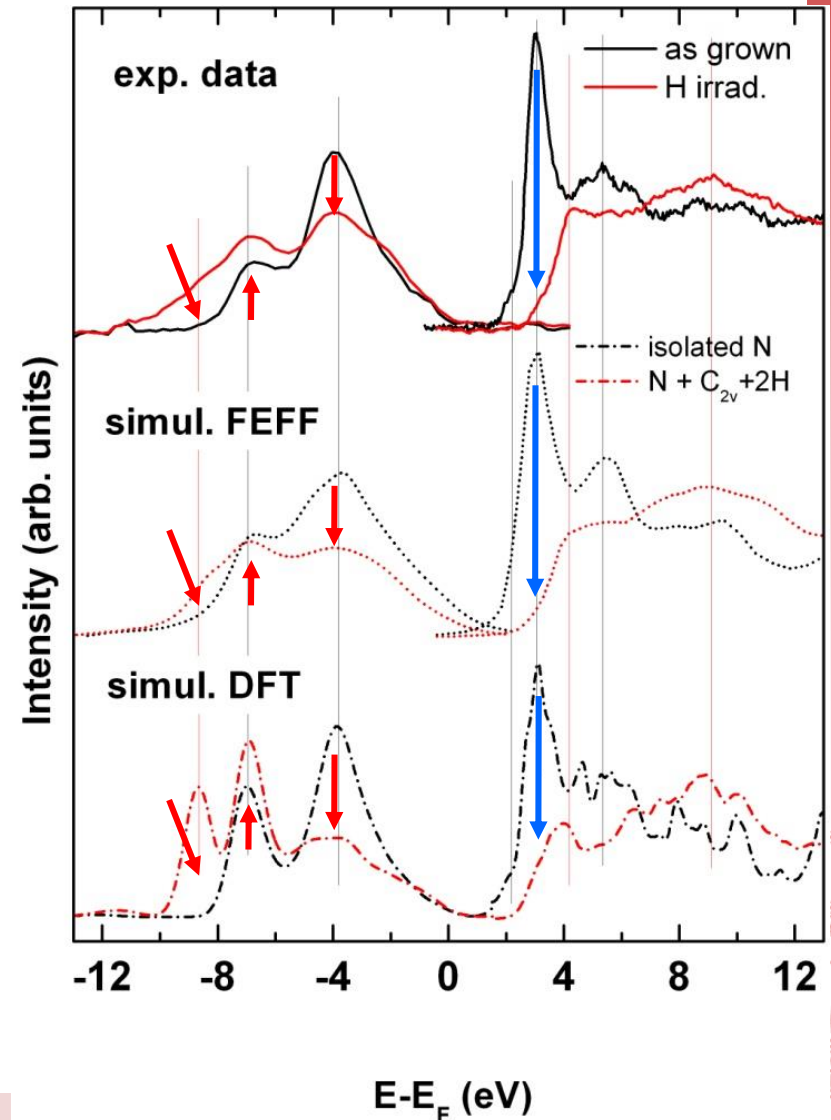


XAFS and XES simulations



Good news: all spectral features are well reproduced by:

- MS spectral simulations based on DFT atomic structure
- ab-initio DFT simulations of electronic and atomic structure



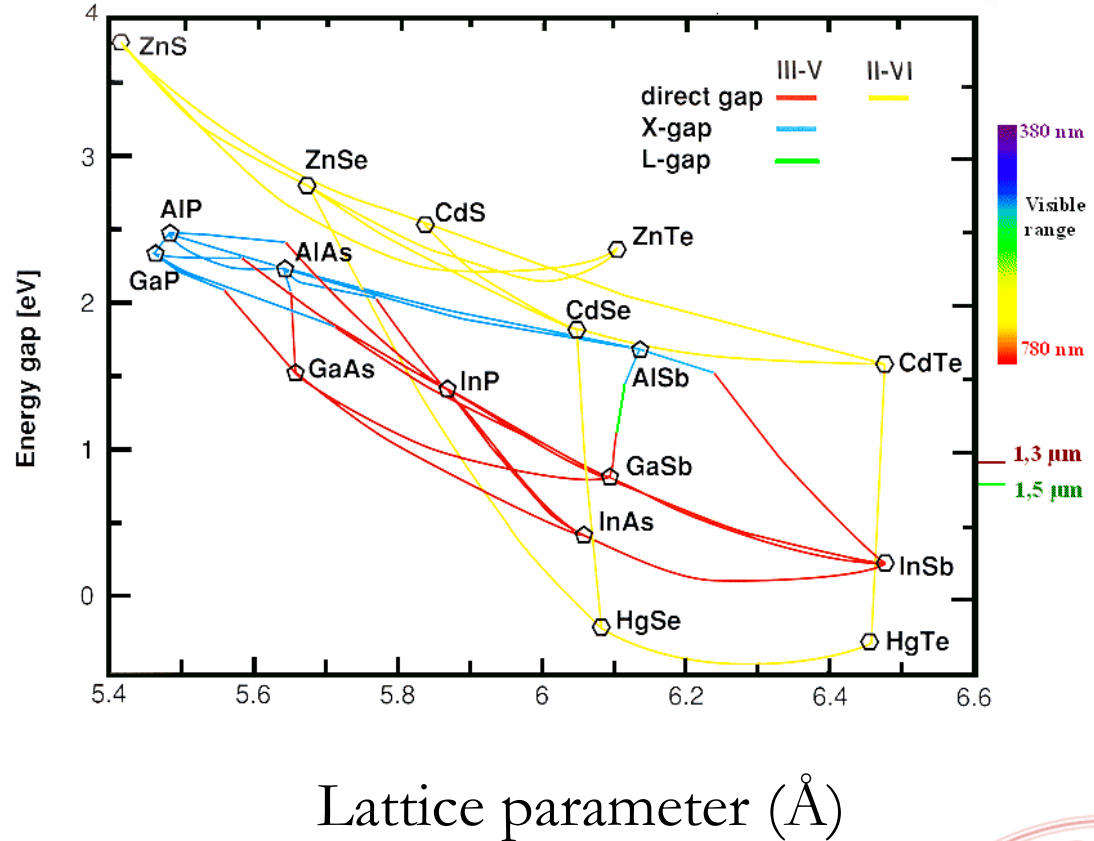
XAFS and alloys

- High resolution in probing the local coordination in first few coordination shells
- Study, as a function of composition
 - Deviation of local structure from average structure
 - Atomic ordering



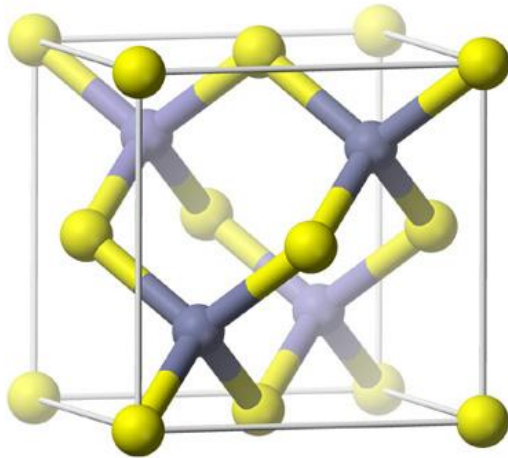
Semiconductor alloys

- For example:
 $\text{In}_x\text{Ga}_{1-x}\text{As}$
- Alloying leads to changes in
 - band-gap
 - lattice parameter

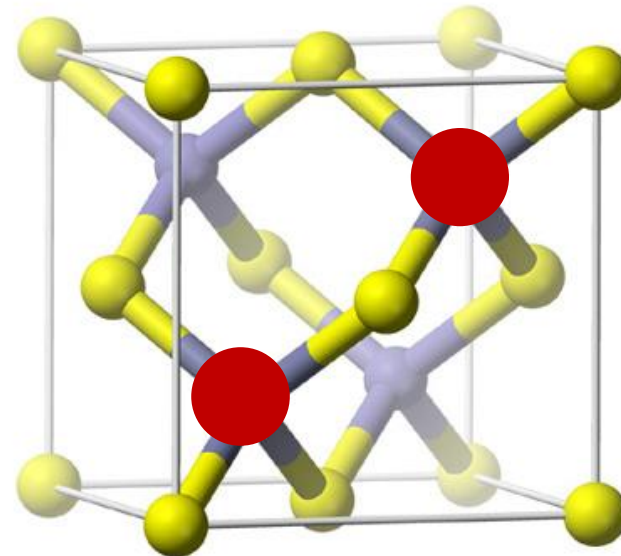


Vegard's law & Virtual Crystal Approximation

- The lattice parameter depends linearly on concentration: “Vegard's law”
- VCA: a linear and isotropic variation of the local structure with concentration
 - All atoms retain symmetric tetrahedral bonding



GaAs

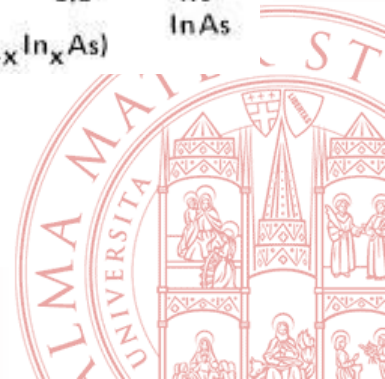
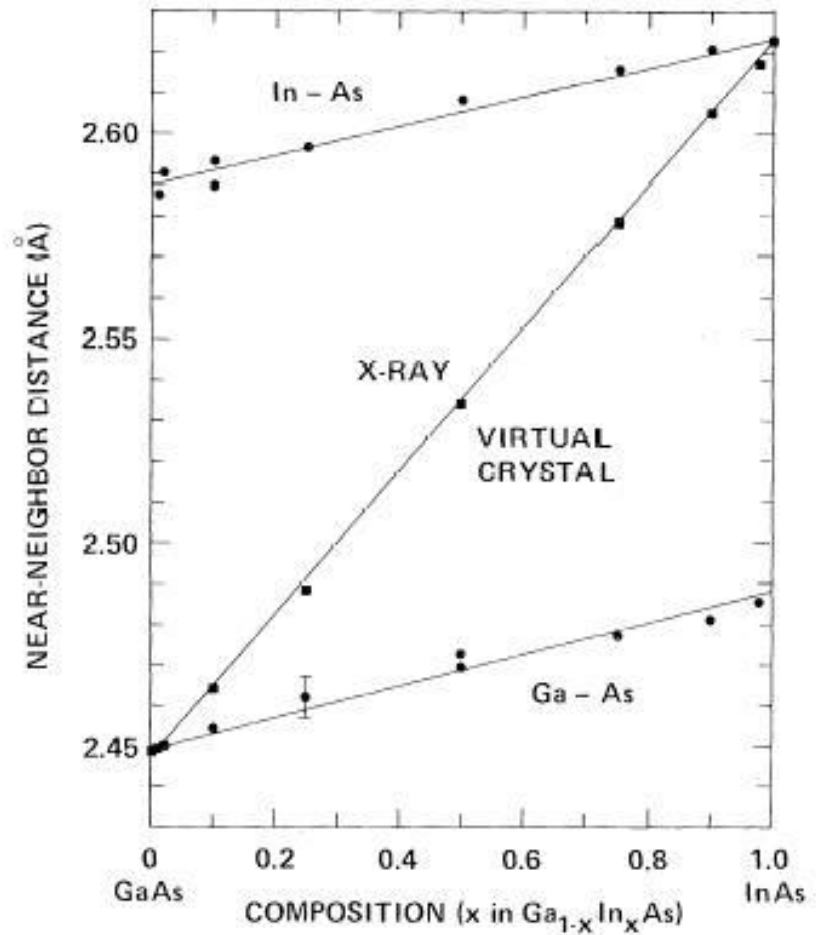


In_{0.5}Ga_{0.5}As



Bond lengths in $\text{In}_x\text{Ga}_{1-x}\text{As}$

- The high resolution of EXAFS in determining bond lengths (0.01 Å) has shown that they stay close to sum of covalent radii
- Violation of the VCA
- First evidence of strong local structural distortions
- Mikkelsen Jr. and Boyce, Phys. Rev. Lett. **49**, 1412 (1982)



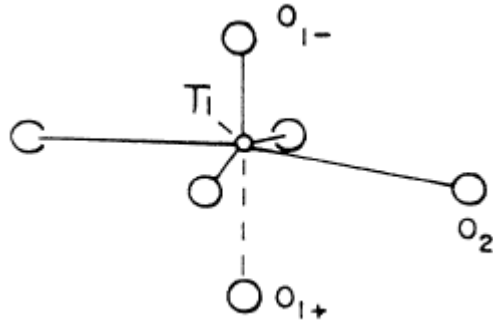
XAFS and phase transitions

- Measure local structure through the phase transition
- XAFS has highlighted the difference between the real local structure and the average structure



Ferroelectric Phase transitions in PbTiO_3

Sicron, Ravel, Yacoby, Stern, Dogan and Stern, Phys. Rev. B 50, 13168 (1994)

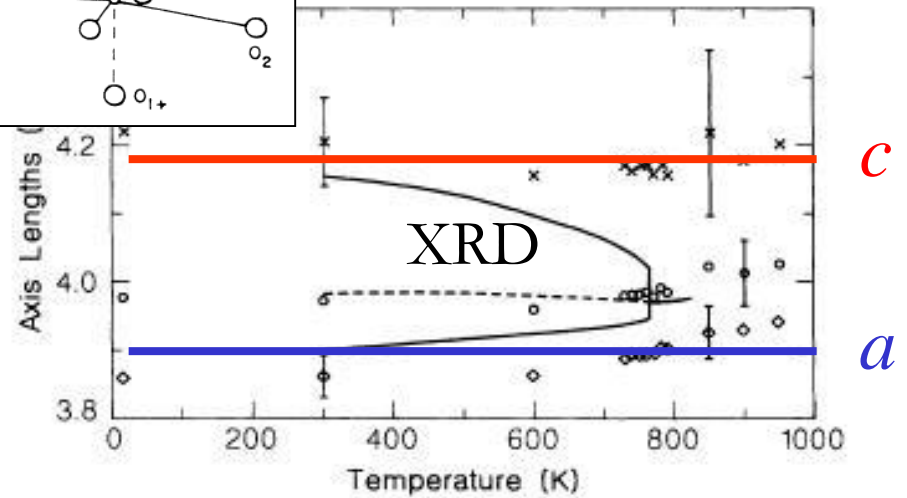
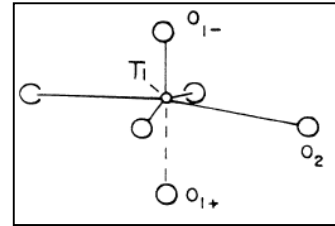


low temp

- At $T_c = 763 \text{ K}$ PbTiO_3 undergoes tetragonal to cubic phase transition
- $T < T_c$ it is ferroelectric (permanent dipole moment)
- Phase transition believed to be purely displacive (no local distortion for $T > T_c$)



Ferroelectric Phase transitions in PbTiO_3



- Ti and Pb XAFS data
- "Local lattice parameters" and local distortions do not change at T_c
 - c : sum of $R(\text{Ti}-\text{O}_1)$
 - a : calculated from $R(\text{Ti}-\text{O}_2)$
- Conclusion
 - local distortions survive above T_c
 - Above T_c random orientation of domains with permanent dipole moment



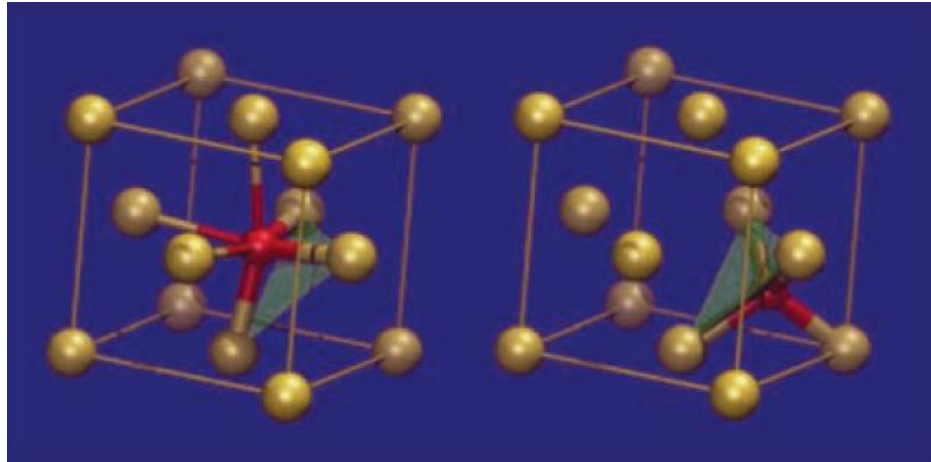
Phase change mechanism in optical media

Kolobov et al., Nature Materials **3**, 703 (2004)

- Phase change optical discs used in DVD-RAMs are based on $\text{Ge}_2\text{Sb}_2\text{Te}_5$ (GST)
- Writing: appropriate laser pulses induce reversible phase changes from amorphous to crystalline
- Reading: the reflectivity of the two phases is different
- What is associated structural change?



Phase change mechanism in optical media



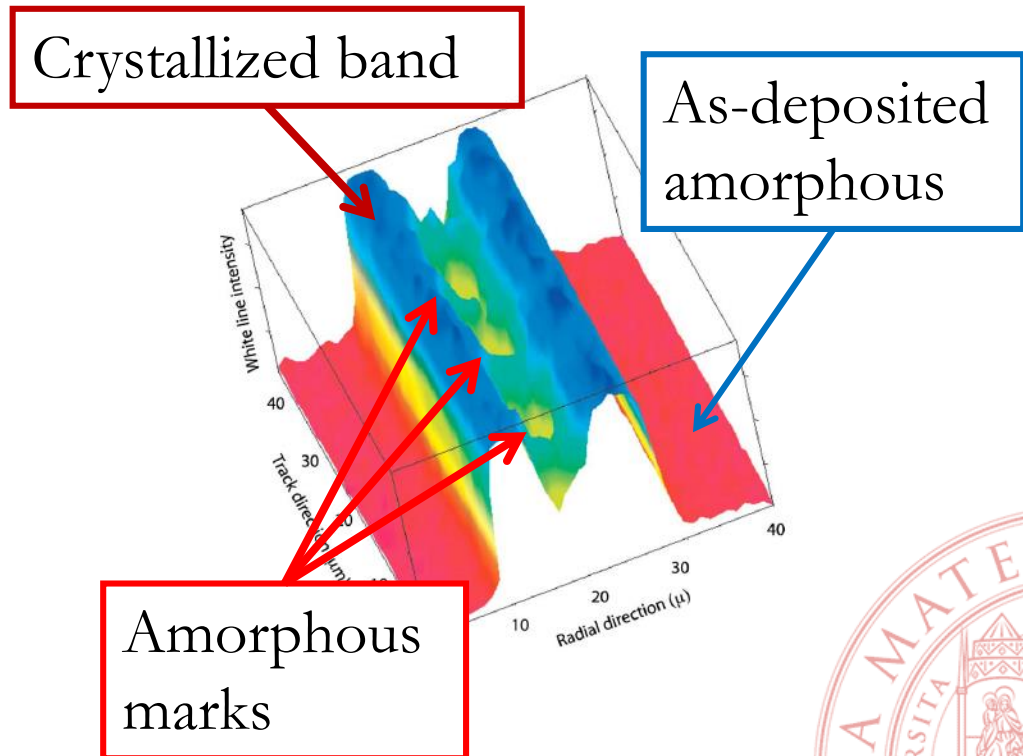
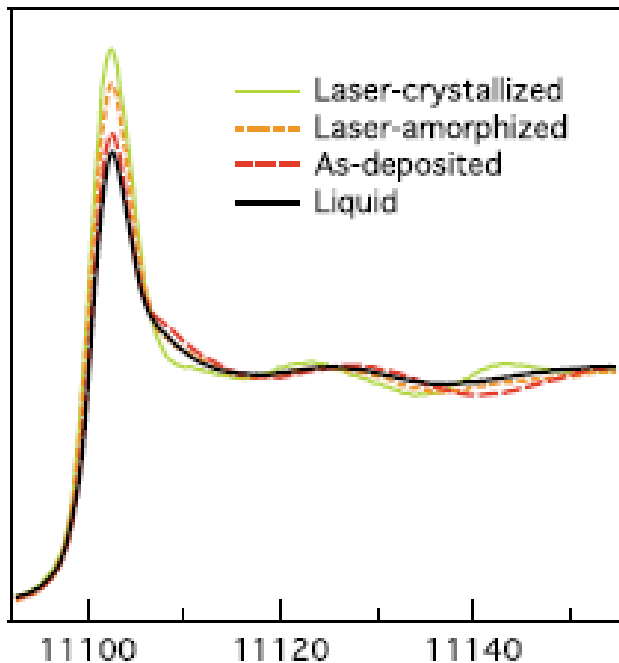
- Phase change is based on “umbrella flip” of Ge, from octahedral to tetrahedral coordination within Te fcc lattice
 - Three strong Ge – Te covalent bonds remain intact
 - Weaker Ge – Te bonds are broken by laser pulse
- Phase change in GST is fast and stable because the process does not require rupture of strong bonds or diffusion



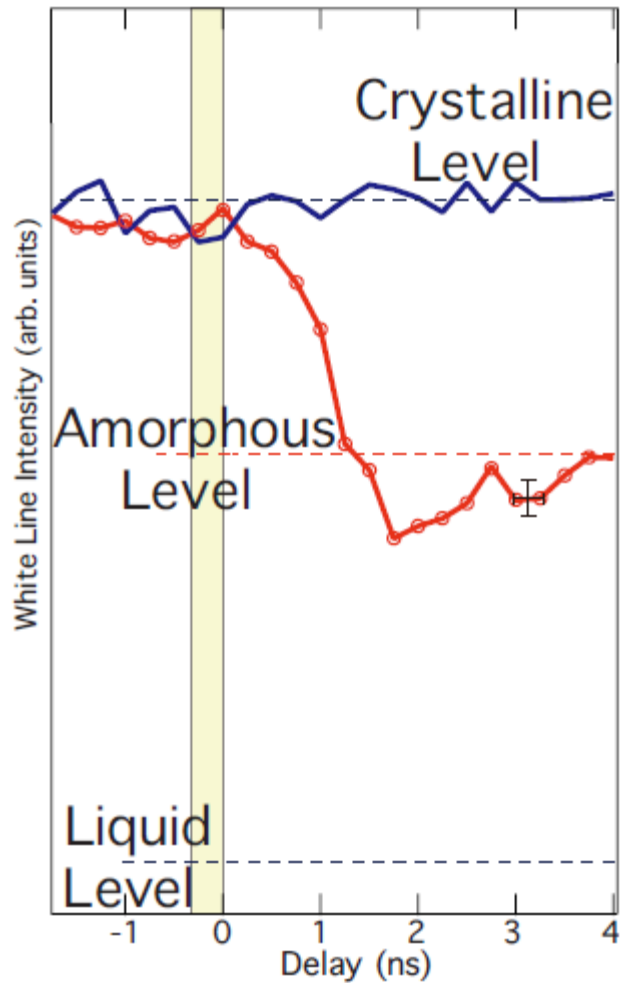
Time resolved XAFS of phase change

Fons et al., Phys. Rev. B **82**, 041203 (2010)

- Sub nanosecond time resolved XAFS with μm spot size at SPring-8
- The intensity of the “white line” is different for crystalline, amorphous and liquid phases



Time resolved XAFS of phase change



- White line intensity versus time
 - 100 ps time resolution
- Phase change does not involve melting



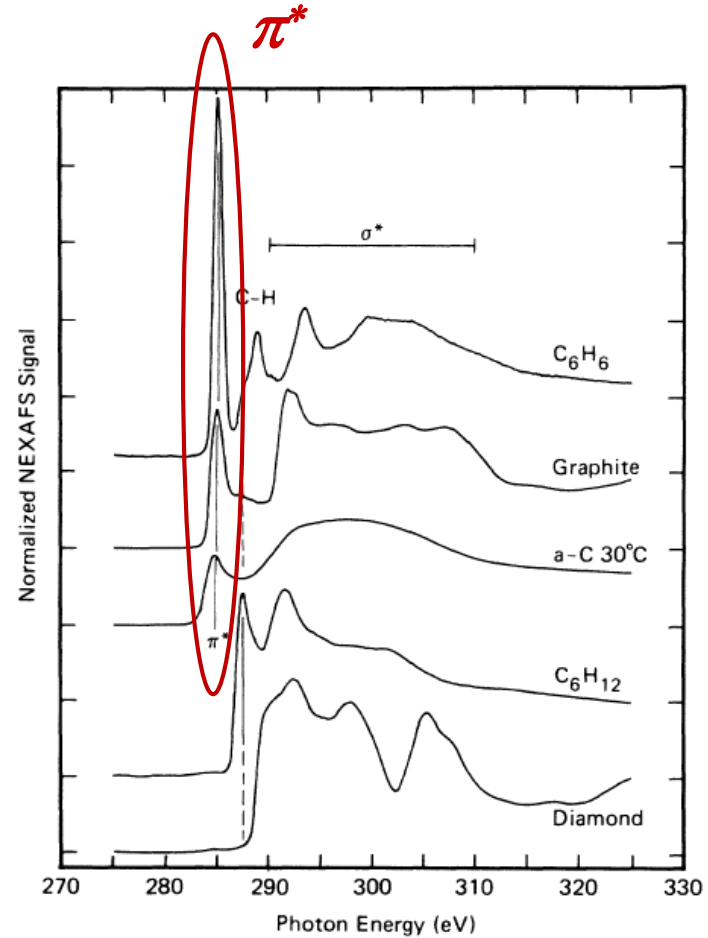
XAFS and thin films / interfaces

- With specific detection schemes sensitivity to very thin films achievable
 - Grazing incidence
 - Electron / fluo detection
- Exploit linear polarization of SR to obtain information on
 - Orientation
 - Lattice symmetry

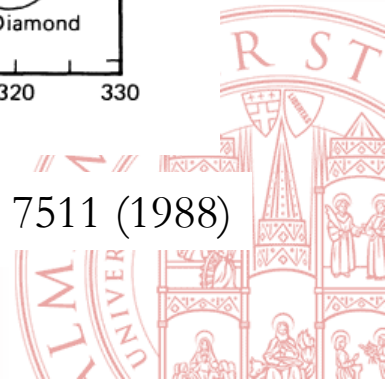


C K edge XANES

- Transitions to π^* molecular orbitals give rise to strong peak

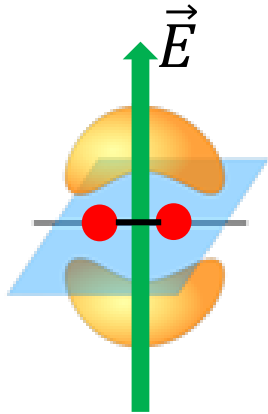


Comelli et al., Phys. Rev. B 38, 7511 (1988)

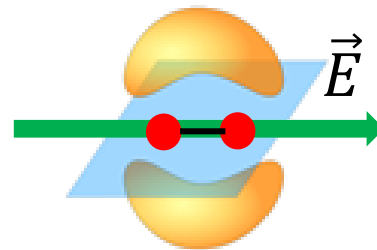


Use of linear dichroism

- Intensity of peaks related to transitions to π^* orbitals depends on the orientation between the orbital and \vec{E}



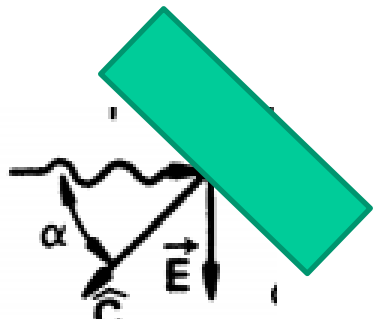
Maximum intensity



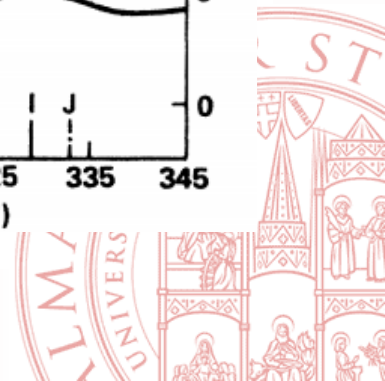
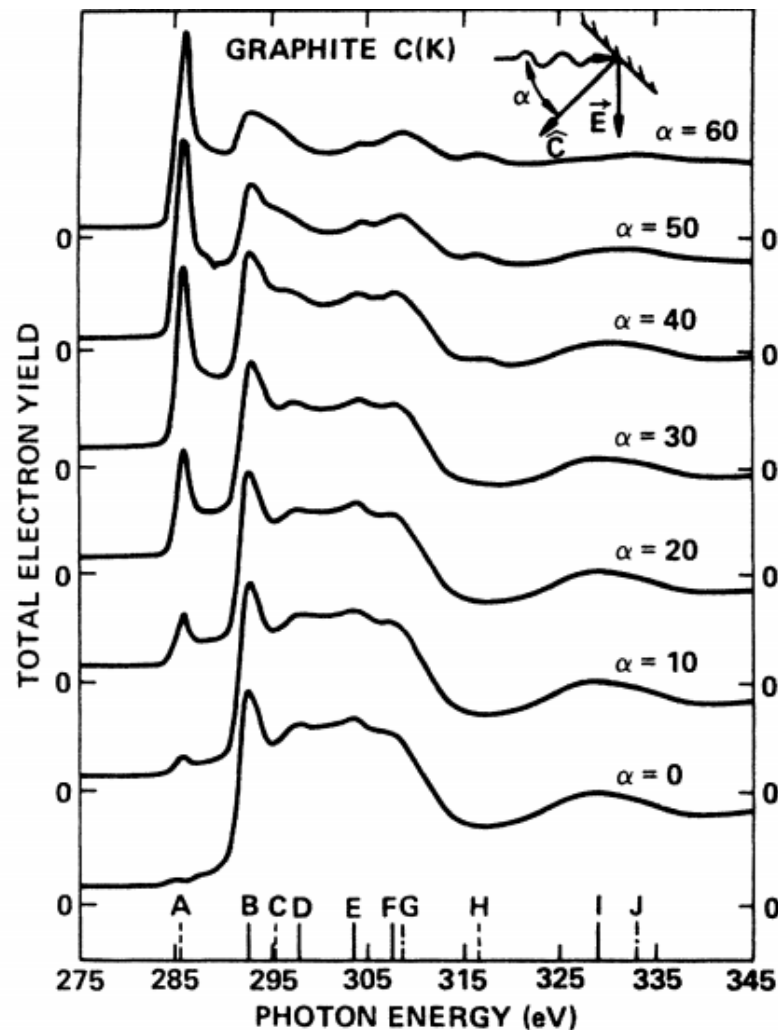
Minimum intensity



C K edge XANES of graphite

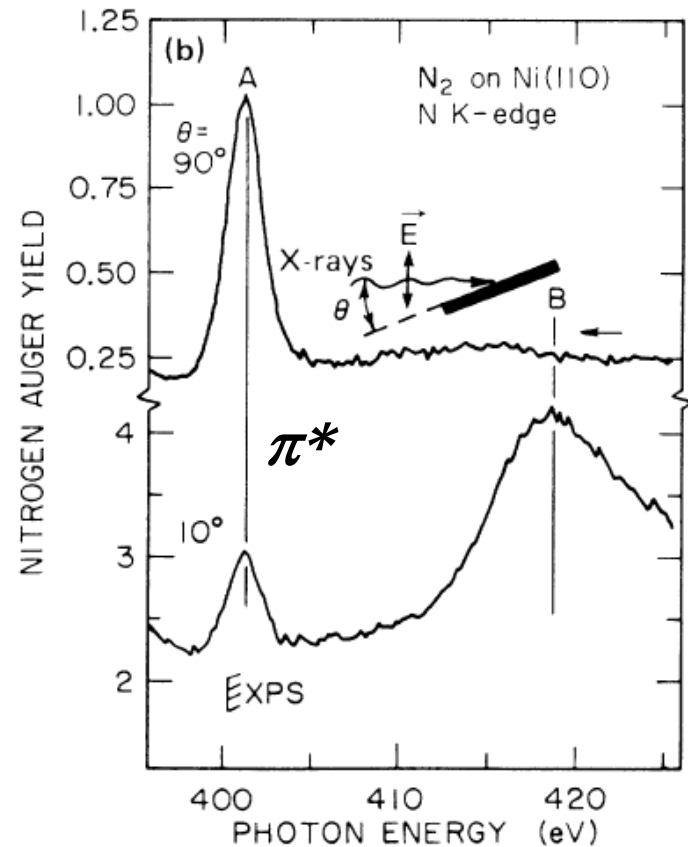
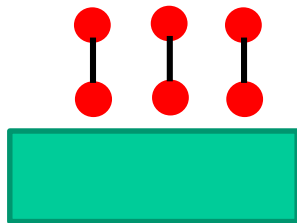


- Very clear dependence of peak due to transitions to π^* orbitals on orientation
- π^* are perpendicular to surface plane
- Rosenberg et al, Phys. Rev. B33, 4034 (1986)



Orientation of molecules on surfaces

- Typical application: determination of the orientation of molecules on single crystal surfaces
- N_2 on Ni(110)
- Molecules are "vertical"



Stöhr & Oukta, Phys. Rev. B 36, 7891 (1987)



XAFS and nanostructures

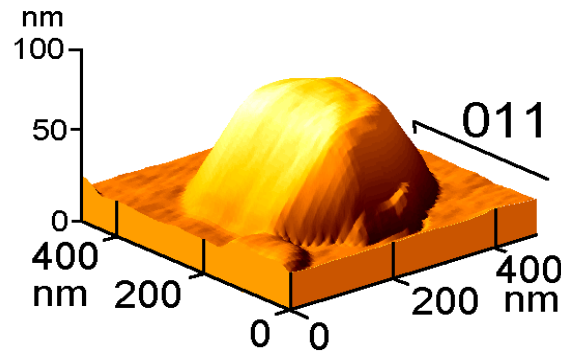
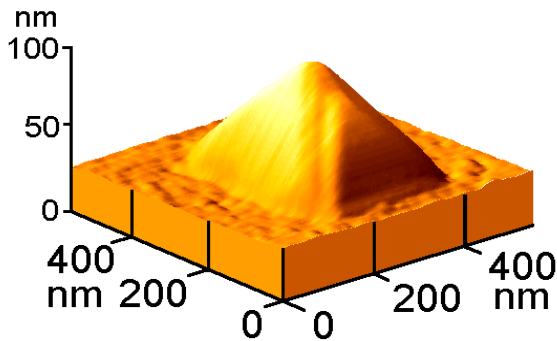
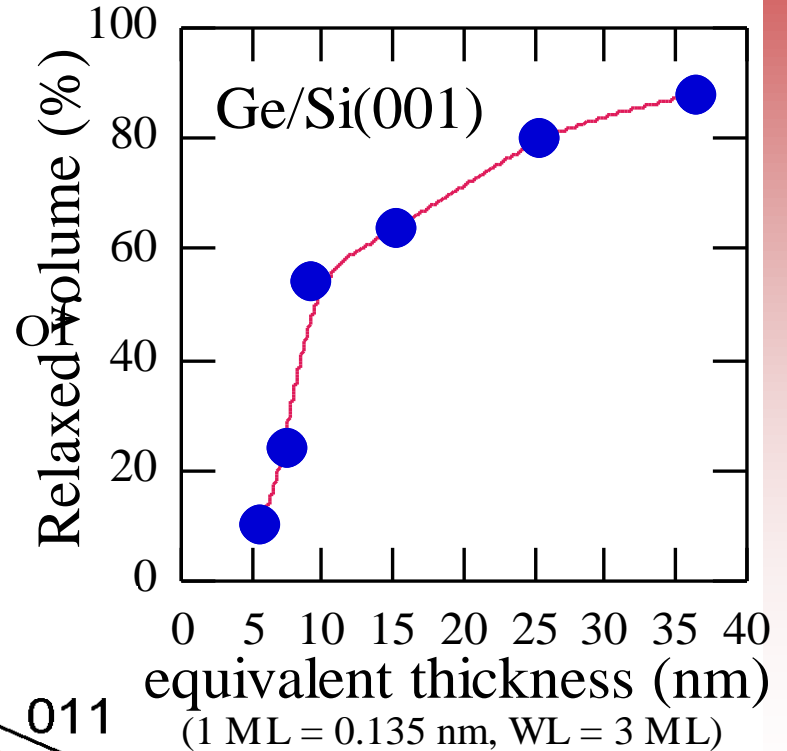
- XAFS is a local, short range, effect
 - Origin: core hole lifetime ($t_{hole} = 10^{-16} - 10^{-15}$ s) and electron mean free path (5 – 10 Å).
- Same formalism applies to molecule, cluster or crystalline solid
 - insensitive to variations of morphology
 - sensitive to low thicknesses, high dilutions
- Excellent probe of **variations** in local environment due to
 - Size effects
 - Change 3D / 2D / 1D



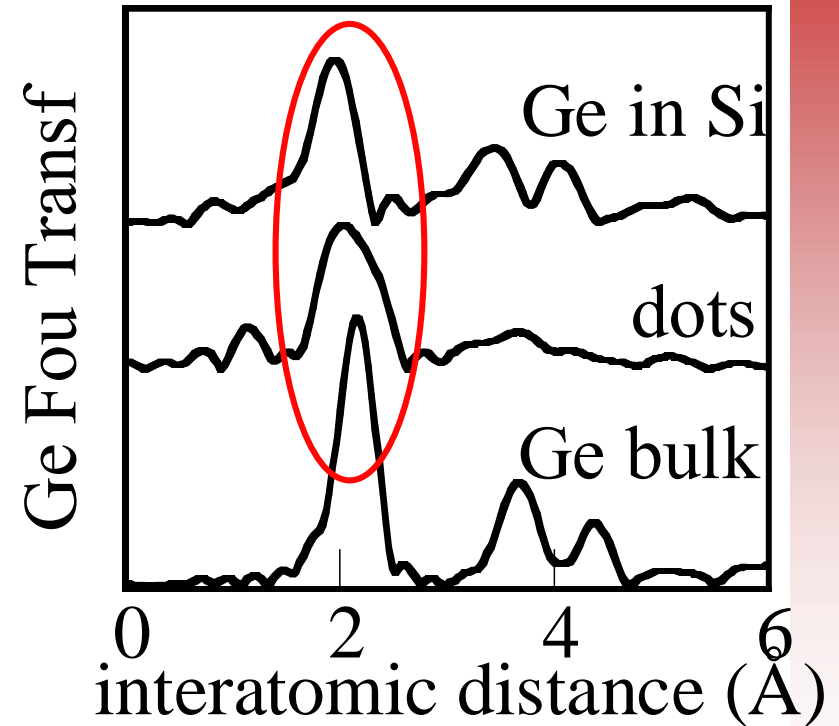
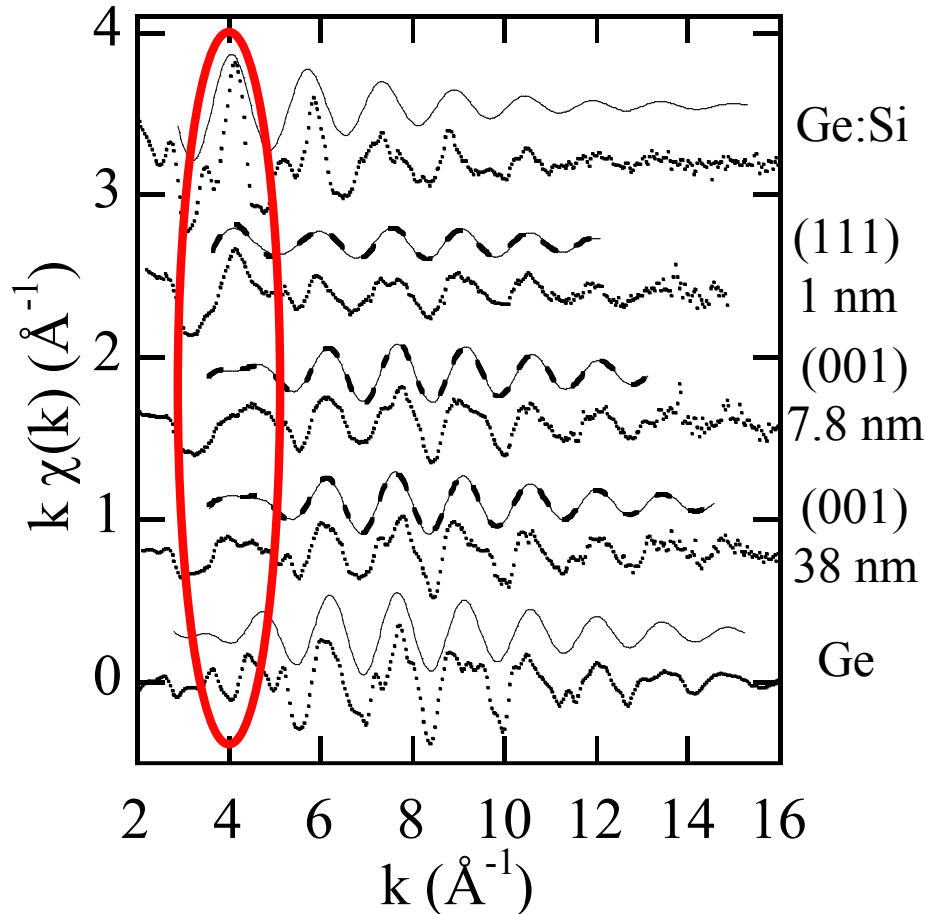
Ge Quantum Dots

F. Boscherini, G. Capellini, L. Di Gaspare, F. Rosei, N. Motta, and S. Mobilio, Appl. Phys. Lett. **76**, 682 (2000)

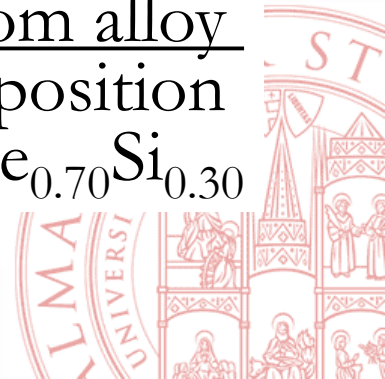
- Need for understanding of local bonding
- Analysis of aspect ratio provides measurement of relative amount relaxed islands



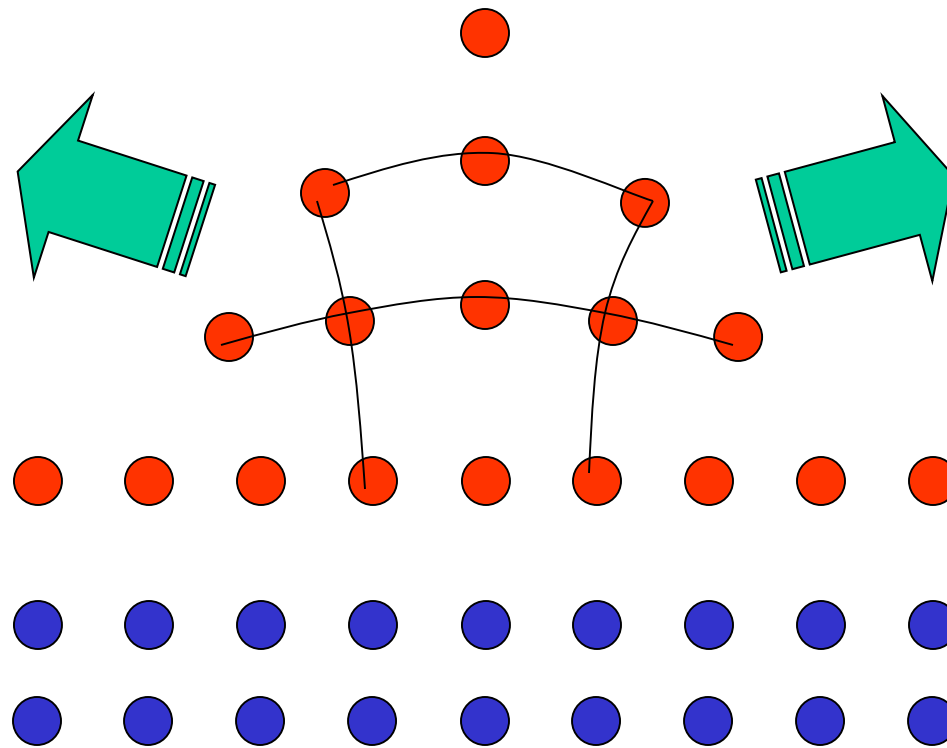
Quantum Dots: Ge edge XAFS



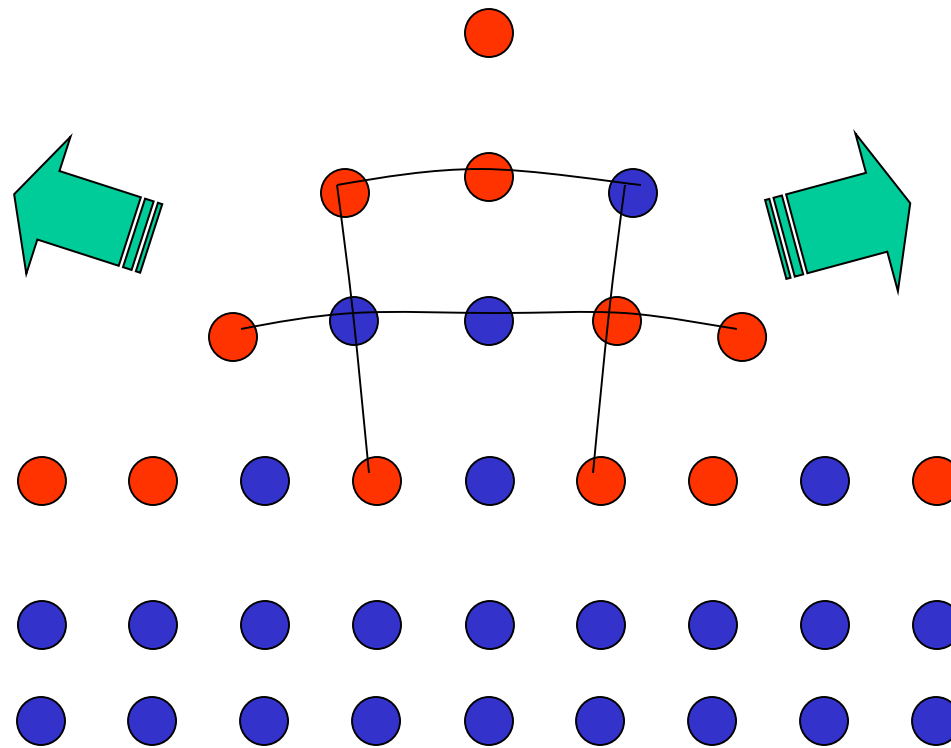
- Assuming random alloy average composition is $\text{Ge}_{0.70}\text{Si}_{0.30}$



Conventional SK growth



SK growth with interdiffusion

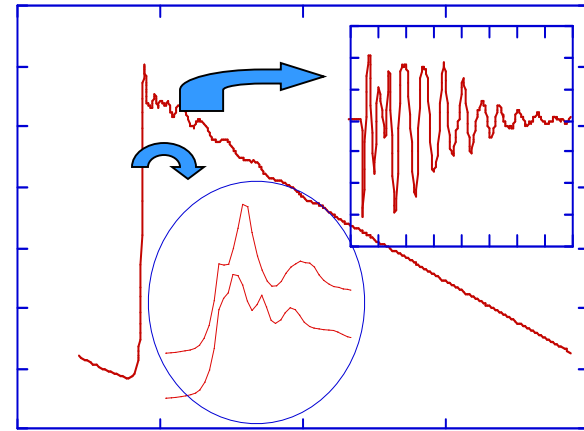
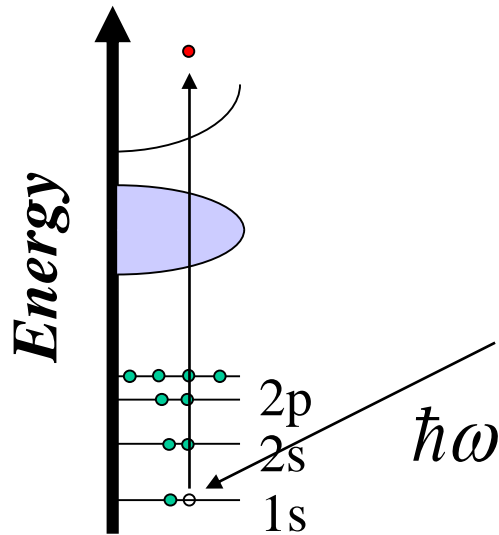


Conclusions

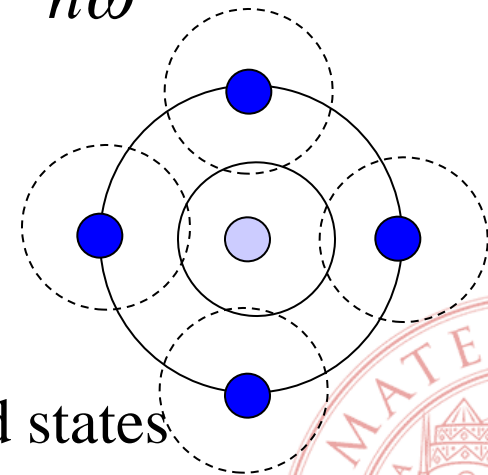
- XAFS has been used to address important structural issues in materials/nano science
- It has specific advantages, especially
 - Atomic selectivity
 - Sensitivity to high dilutions & surfaces/interfaces
 - Equally applicable to ordered or disordered matter
 - EXAFS: high resolution for first few coordination shells
 - XANES:
 - valence/oxidation state
 - 3D structural sensitivity
 - μm spot size now available and decreasing fast
 - Time resolution in the 10 -100 ps range available and with FELs decreasing to 10 fs



X-Ray Absorption Fine Structure



$\hbar\omega$



- “EXAFS”: Coordination numbers
Interatomic distances
Disorder of distances
- “XANES”: Absorber symmetry and valence/oxidation state
Electronic structure of unoccupied states
Medium range structure

EXAFS

- Extended X-ray Absorption Fine Structure
- When applicable, fit with the “standard” EXAFS equation

From *ab-initio* calculations or from reference compounds

$$\chi(k) = S_0^2 \sum_{j=\text{shells}} N_j A_j(k) \text{Sin}[2kr_j + \overbrace{\varphi_j + 2\delta_1}] e^{-2k^2\sigma_j^2}$$

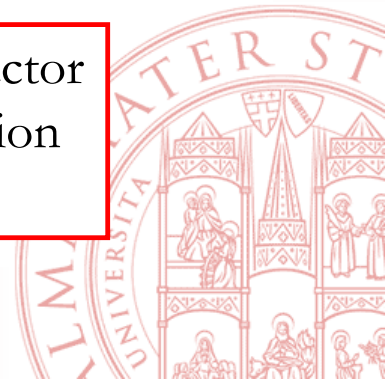
$$k = \frac{\sqrt{2m(\hbar\omega - E_B)}}{\hbar}$$

Measure:

Coordination
number

Interatomic
distance

Debye Waller factor
- thermal vibration
- static disorder



XANES

- X-ray Absorption Near Edge Structure
(also NEXAFS)

$$\sigma(\hbar\omega) = 4\pi^2 \alpha \hbar\omega \left| \langle i | \hat{\epsilon} \cdot \vec{r} | f \rangle \right|^2 \rho(E_f) \quad \Delta l = \pm 1, \Delta m_\ell = 0$$

(lin. pol. light)

- “Molecular orbital” approach: 1 electron approximation, constant matrix element: probe **site** and **symmetry** projected density of states of final electronic states
- “Multiple scattering” approach: structural interpretation through simulation



Characteristics of XAFS

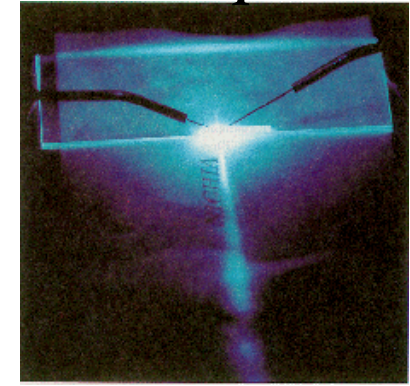
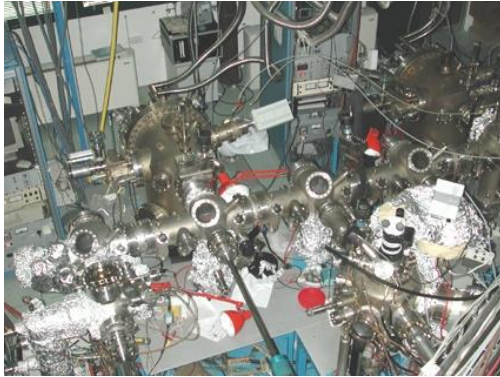
- Atomic selectivity (choose Z via photon energy)
- Equally applicable to ordered or disordered matter
- A core level technique: a local probe
- Interesting underlying physics
- Sensitive to high dilutions
- EXAFS: high distance resolution
- XANES: 3D structural sensitivity
- Recent developments:
 - Sub μm spot size
 - ns, ps and ...fs time resolution



Role of XAFS in Materials Science

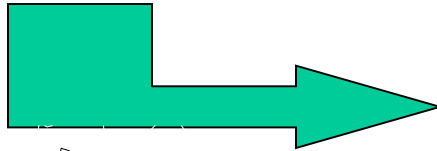
Growth

Physical Properties

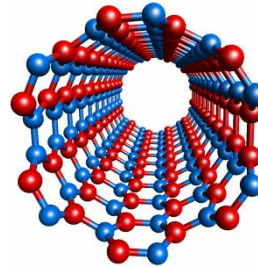
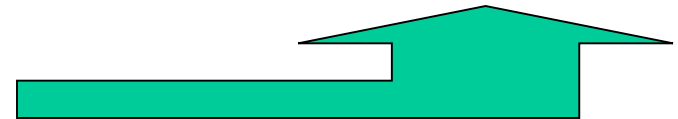


Structure

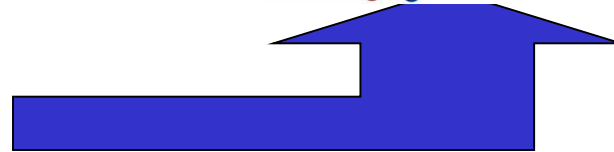
MBE@TASC



Nakamura et al., Jpn. J. Appl. Phys. 35, L217, 1996



XAFS



Objective: an understanding of physical properties of novel materials based on knowledge of their local structure

